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Preparation and photocatalytic activity of mesoporous anatase $TiO₂$ nanofibers by a hydrothermal method

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Abstract

Mesoporous anatase TiO₂ nanofibers were prepared by a simple hydrothermal post-treatment of titanate nanotubes. The as-prepared samples were characterized with transmission electron microscopy (TEM), thermogravimetry (TG), X-ray diffraction (XRD), field emission scanning electron microscope (FESEM), and N_2 adsorption–desorption measurement. The photocatalytic activity of the mesoporous TiO₂ nanofibers was evaluated by photocatalytic oxidation of acetone in air. The effects of hydrothermal post-treatment time on the structures and photocatalytic activity of the mesoporous TiO₂ nanofibers were discussed. The results showed that the photocatalytic activity of the TiO₂ nanofibers prepared by this method exceeded that of Degussa P25 when the hydrothermal post-treatment time was kept at 200 ℃ for 3–24 h. This could be attributed to the fact that the former had smaller crystallite size, larger specific surface area, and higher pore volume. © 2006 Elsevier B.V. All rights reserved.

Keywords: Mesoporous; Anatase nanofibers; Photocatalytic activity; Hydrothermal post-treatment

1. Introduction

Many efforts have been devoted in recent years to developing titania semiconductor photocatalyst with highly photocatalytic activity for environmental protection procedures such as air purification, water disinfection, and hazardous water remediation due to its strong oxidizing power, chemical inertness, and nontoxicity $[1-5]$. It is well known that the performance of TiO₂ depends strongly on its crystallite phase, dimension, and morphology, since they give a decisive influence on the chemical and physical properties of $TiO₂$. One-dimensional (1D) nanostructured materials, such as nanotubes and nanofibers, are of particular significance due to their superior physicochemical properties, such as remarkable electronic, magnetic, optical, catalytic, and mechanical properties, and potential applications in environmental purification, nanodevices, gas sensors, and higheffect solar cell [\[6–12\].](#page-6-0) For example, several recent studies indicated that $TiO₂$ nanotubes possessed better properties in photocatalysis [\[13,14\]](#page-6-0) and sensing [\[15\],](#page-6-0) in comparison to colloidal or other forms of $TiO₂$. Moreover, it is expected that the com-

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bination of the 1D nanostructures and mesoporous structures would endow titania with unique properties and multiple functions. However, much attention was mainly paid to mesoporous $TiO₂$ particles or thin films, while 1D mesoporous $TiO₂$ materials were less studied, mainly due to the difficulties in fabricating such structures with good control over their dimensions [\[16–20\].](#page-6-0) Therefore, it is interesting to explore the synthesis of 1D nanostructured mesoporous $TiO₂$ materials and their photocatalytic activity.

Recently, many attempts have been made to prepare 1D TiO2 and/or titanate nanostructured materials with large specific surface area and high pore volume $[6-10,21-23]$. Crystalline TiO₂ nanotubes or nanofibers were usually synthesized using porous anodic alumina as templates [\[16,18,19,24,25\].](#page-6-0) However, the prepared nanotubes or nanofibers by the above method generally had large diameters (>50 nm) due to the confinement of the molds used. Using a simple hydrothermal treatment of crystalline $TiO₂$ particles with NaOH aqueous solutions, highquality nanotubes with uniform diameter of around 10 nm were obtained and their specific surface area reached more than $400.0 \,\mathrm{m}^2/\mathrm{g}$ [\[6,7\]. U](#page-6-0)nfortunately, the obtained nanotubes showed almost no photocatalytic activity for the photocatalytic degradation of acetone in our experiment. Following the pioneering works, the nanotubes obtained by this method were actually not

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 $TiO₂$, but might be of protonic titanate [\[9,26\]. C](#page-6-0)onsidering their large specific surface area, high pore volume, and unique morphology, the obtained nanotubes will offer another possibility to design various $TiO₂$ -related materials by post-treatment methods, such as hydrothermal post-treatment and well-controlled calcination.

Hydrothermal synthesis, a low temperature technology for materials development, was widely applied to prepare various materials due to high reactivity of reactants, easy control of solution, less air pollution, and low energy consumption under hydrothermal conditions [\[27\].](#page-6-0) In the present work, titanate nanotubes were first prepared by a hydrothermal reaction using a 10 M NaOH aqueous solution and Degussa P25 as precursors. Subsequently, the as-prepared nanotubes were hydrothermally post-treated at 200 ◦C for 1–24 h, leading to the formation of mesoporous anatase $TiO₂$ nanofibers. The photocatalytic activity of the mesoporous $TiO₂$ nanofibers was evaluated by photocatalytic oxidation of acetone in air. The effects of hydrothermal post-treatment time on the phase structure, crystallization, crystallite size, morphology, specific surface area, pore structures, and photocatalytic activity of the mesoporous $TiO₂$ nanofibers were discussed. This is the first report on the preparation of mesoporous anatase $TiO₂$ nanofibers by the hydrothermal post-treatment of the titanate nanotubes. This work may provide new insights into the preparation of mesoporous $TiO₂$ photocatalyst with highly photocatalytic activity.

2. Experimental

2.1. Preparation of titanate nanotubes

Titanate nanotubes were prepared using a chemical process similar to that described by Kasuga et al. [\[6,7\]](#page-6-0) and Tian et al. $[28]$. TiO₂ source used for the titanate nanotubes was commercial-grade TiO₂ powder (P25, Degussa AG, Germany) with crystalline structure of ca. 20% rutile and ca. 80% anatase and primary particle size of ca. 30 nm. In a typical nanotube preparation, $1.5 g$ of the TiO₂ powder was mixed with $140 ml$ of 10 M NaOH solution, followed by hydrothermal treatment of the mixture at 150° C in a 200 ml Teflon-lined autoclave for 48 h. After hydrothermal reaction, the precipitate was separated by filtration and washed with a 0.1 M HCl solution and distilled water until the pH value of the rinsing solution reached ca. 6.5, approaching the pH value of the distilled water. The washed samples were dried in a vacuum oven at 80° C for 8 h.

2.2. Preparation of mesoporous TiO2 nanofibers

In the preparation of mesoporous $TiO₂$ nanofibers, 0.55 g of the as-prepared titanate nanotubes were mixed with 80 ml of distilled water, followed by hydrothermal post-treatment of the mixture at $200\degree C$ in a 100 ml Teflon-lined autoclave for 1–24 h. After the hydrothermal post-treatment, the obtained samples were filtered and dried in a vacuum oven at 80° C for 8 h.

2.3. Characterization

Transmission electron microscopy (TEM) analyses were performed with a JEOL JEM 2010FEF electron microscope, using 200 kV accelerating voltage. The X-ray diffraction (XRD) patterns were obtained on a Bruker D8 Advance X-ray diffractometer using Cu K α radiation at a scan rate of 0.05° 2θ s⁻¹ and were used to determine the identity of crystalline phase and the crystallite size. The accelerating voltage and the applied current were 40 kV and 40 mA, respectively. The surface morphologies of the samples were observed using a JEOL-6700F field emission scanning electron microscope (FESEM). The thermogravimetry (TG) was performed on a TG-DTA instrument (Setaram TG-DTA 92-16, France), under a heating rate of 10 K min−¹ from room temperature to 1000 ◦C. Nitrogen adsorption–desorption isotherms were obtained on an AUTOSORB-1 (Quantachrome Instruments, USA) nitrogen adsorption apparatus. All the samples were degassed at 80 ◦C prior to BET measurements. The Brunauer–Emmett–Teller (BET) specific surface area (S_{BET}) was determined by a multipoint BET method using the adsorption data in the relative pressure (P/P_0) range of 0.05–0.3. The desorption isotherm was used to determine the pore size distribution using the Barret–Joyner–Halender (BJH) method [\[29\].](#page-6-0) The nitrogen adsorption volume at the relative pressure (P/P_0) of 0.994 was used to determine the pore volume and the average pore size.

2.4. Measurement of photocatalytic activity

Acetone (CH_3COCH_3) is a common chemical that is used extensively in various industrial and domestic applications. Therefore, we chose it as a model contaminate chemical. Photocatalytic oxidation of acetone is based on the following reaction [\[30–33\]:](#page-6-0)

$$
CH3COCH3 + 4O2 \rightarrow 3CO2 + 3H2O
$$
 (1)

The measurement of photocatalytic activity of the mesoporous $TiO₂$ nanofibers was performed in a 15 L reaction reactor using the photodegradation of acetone with an initial concentration of 350 ± 20 ppm. The detailed experimental setup and process have been reported elsewhere [\[33\].](#page-6-0) The $TiO₂$ photocatalysts were prepared by coating an aqueous suspension of $TiO₂$ samples onto three dishes with a diameter of ca. 7.0 cm. The dishes containing catalysts were dried at 100 ◦C and then cooled to room temperature before being used. The weight of the $TiO₂$ catalysts used for each experiment was kept at ca. 0.5 g. After the catalysts were placed in the reactor, a small amount of acetone was injected with a syringe into the reactor. The reactor was connected to a $CaCl₂$ -containing dryer used for controlling the initial humidity in the reactor. The acetone vapor was allowed to reach adsorption–desorption equilibrium with catalysts in the reactor prior to UV light irradiation. Integrated UV intensity in the range of 310–400 nm striking the coatings, measured with a UV radiometer (Model: UV-A, made in Photoelectric Instrument Factory of Beijing Normal University), was 2.5 mW/cm^2 , while the peak wavelength of UV light was 365 nm. The concentration analysis of acetone, carbon dioxide, and water vapor in the reactor was conducted on line with a Photoacoustic IR Multigas Monitor (INNOVA Air Tech Instruments Model 1312). The photocatalytic activity of the $TiO₂$ samples can be quantitatively evaluated by comparing the apparent reaction rate constants. The photocatalytic oxidation of acetone is a pseudo-first-order reaction and its kinetics may be expressed as follows: $ln(C_0/C) = kt$ [\[31,32\],](#page-6-0) where k is the apparent reaction rate constant, C_0 and *C* are the initial concentration and the reaction concentration of acetone, respectively.

3. Results and discussion

3.1. Morphology and structure of titanate nanotubes

A large amount of nanotubes with diameters of 7–12 nm and lengths of several hundreds nanometers were prepared by a hydrothermal reaction between Degussa P25 and a 10 M NaOH aqueous solution at 150° C for 48 h (as shown in Fig. 1). EDX analysis demonstrated the absence of sodium ions in the obtained nanotubes (not shown here). Therefore, it could be concluded that the sodium ions were substituted completely by protons after the nanotubes were washed with a HCl aqueous solution and distilled water. Fig. 2 depicts the XRD patterns of P25 powder and the obtained nanotubes. The precursor P25 showed the presence of both anatase and rutile phases. After hydrothermal treatment of P25 in a 10 M NaOH solution for 48 h, the pattern of the resulted powders (nanotubes) could not correspond to anatase, rutile, brookite or their mixtures. The crystal structure of the nanotubes was similar to that of $H_2Ti_3O_7$ (Na₂T_{i3}O₇) [\[8,9\],](#page-6-0) Na*x*H2−*^x*Ti3O7 [\[26\],](#page-6-0) H*x*Ti2−*x*/4*x*/4O4 (*x* = 0.75) [\[34\],](#page-6-0) Na2Ti2O4(OH)2 [\[35\],](#page-6-0) or Na*y*H2−*y*Ti*n*O2*ⁿ*+1·*x*H2O [\[36\],](#page-6-0) probably due to their same layered titanate family. TG curve (not shown here) showed that a mass loss of ca. 20.1% was observed in this study after the nanotubes were heated from room temperature to 1000 ◦C. Considering the absence of Na in the nanotubes and a large mass loss, the obtained nanotubes may be described as $H_2Ti_nO_{2n+1}.xH_2O$ and can be attributed to protonic titanate.

Fig. 1. TEM image of the titanate nanotubes obtained from hydrothermal treatment of P25 with NaOH solution at 150 ◦C for 48 h.

Fig. 2. XRD patterns of the raw material (P25) and the titanate nanotubes.

3.2. Phase structure and surface morphology of mesoporous TiO2 nanofibers

Fig. 3 shows the XRD patterns of the $TiO₂$ nanofibers obtained by hydrothermal post-treatment of the titanate nanotubes at $200\degree$ C for 1, 3, 7, 12, and 24 h. After hydrothermal post-treatment for 1 h, the intensity of diffraction peak located at ca. 11◦ (nanotubes) was weakened significantly and the anatase peak at ca. 25.5◦ appeared, indicating the destruction of titanate structure and the formation of anatase phase. It should be noted that the titanate and anatase $TiO₂$ possess some similar structure features. In titanium oxide lattice, a $TiO₆$ octahedron in anatase phase shares four edges with other octahedrons to form a zigzag ribbon structure [\[37\]. I](#page-6-0)n addition to the four edge-sharing $TiO₆$ octahedras and the zigzag ribbons structure in the titanate lattice, these zigzag ribbons then form a titanate layer by sharing the corners of these ribbons, and the protons or sodium ions can exist between the titanate layers [\[38\].](#page-6-0) Therefore, it seems that the anatase phase was formed by the dehydration among the layers and the rearrangement of the zigzag ribbon structures. With increasing hydrothermal post-treatment time, the diffrac-

Fig. 3. XRD patterns of the $TiO₂$ nanofibers obtained from hydrothermal posttreatment of the titanate nanotubes at $200\degree$ C for 1, 3, 7, 12, and 24 h.

tion peaks of the titanate nanotubes disappeared and the peaks intensities of anatase phase increased ([Fig. 3\),](#page-2-0) indicating the enhancement of crystallization in anatase phase. Fig. 4 shows the average crystallite size of the $TiO₂$ nanofibers as a function of the hydrothermal post-treatment time. With increasing hydrothermal post-treatment time, the average crystallite size increased from 8.1 to 27.3 nm due to the growth of $TiO₂$ crystallites. However, the crystallite size of all the $TiO₂$ nanofibers was smaller than that of the precursor P25 (ca. 30.0 nm).

Fig. 5 shows the SEM images of the $TiO₂$ nanofibers obtained by hydrothermal post-treatment of the titanate nanotubes at 200 \degree C for 3, 7, and 24 h. After hydrothermal post-treatment for 3 h, the $TiO₂$ nanofibers were observed and exhibited a diameter of $25-50 \text{ nm}$ and a length of 0.2-3 μ m (Fig. 5a and b). Meanwhile, these $TiO₂$ nanofibers tended to aggregate to form nanofiber bundles with diameters of 100–500 nm (Fig. 5a). Further observation indicated that the $TiO₂$ nanofibers were composed of many small $TiO₂$ particles with diameters of 15–30 nm. With increasing hydrothermal post-treatment time

Fig. 4. Effect of hydrothermal treatment time on the crystalline size of $TiO₂$ nanofibers obtained from hydrothermal post-treatment of the titanate nanotubes.

Fig. 5. SEM images of the TiO₂ nanofibers obtained from hydrothermal post-treatment of the titanate nanotubes at 200 ℃ for 3 h (a and b); 7 h (c and d); 24 h (e).

Fig. 6. Nitrogen adsorption–desorption isotherms (a) and pore-size distribution curves (b) of the TiO₂ nanofibers obtained from hydrothermal post-treatment of the titanate nanotubes at 200° C for 0, 1, 7, and 24 h.

from 3 to 7 h, as shown in [Fig. 5c](#page-3-0) and d, the length of the $TiO₂$ nanofibers decreased to \sim 2 µm, while the corresponding diameters increased to 100–500 nm which was equal to the diameters of the $TiO₂$ nanofiber bundles found in [Fig. 5a](#page-3-0). It seems that the hydrothermal dehydration reaction took place in the formed fiber bundlers during the hydrothermal post-treatment (3–7 h), resulting in the formation of $TiO₂$ fibers with larger diameters (100–500 nm). On the other hand, the larger $TiO₂$ fibers were composed of many $TiO₂$ particles with larger diameter of $25-40$ nm [\(Fig. 5c](#page-3-0) and d) in comparison to that of TiO₂ parti-cles found in [Fig. 5b](#page-3-0), indicating the growth of $TiO₂$ particles. After hydrothermal post-treatment of the titanate nanotubes at 200 °C for 24 h, the length of the TiO₂ fibers further become shorter (∼800 nm) and their diameters decreased to less than 100 nm [\(Fig. 5e\)](#page-3-0). Moreover, many TiO₂ particles with diameter of 70–150 nm were observed. It is possible that the further hydrothermal post-treatment at high-temperature (200 °C) and high-pressure conditions caused the destruction of the $TiO₂$ fibers and led to the formation of $TiO₂$ particles.

3.3. BET specific surface areas and pore structures

The nitrogen adsorption–desorption isotherms of the obtained samples are presented in Fig. 6a. All samples showed the type IV isotherms with type H3 hysteresis loops according to BDDT classification [\[29\], i](#page-6-0)ndicating the presence of mesopores (2–50 nm). As for the titanate nanotubes, the observed hysteresis loops shifted to a high relative pressure $P/P_0 \approx 1$, suggesting the presence of large pores (>50 nm)[\[39\]. C](#page-6-0)onsidering the morphology of the nanotubes observed in [Fig. 1, t](#page-2-0)he smaller pores may correspond to the pores inside the nanotubes and the diameter of these pores is equal to the inner diameter of the nanotubes, while the larger pores can be attributed to the aggregation of the nanotubes [\[39\].](#page-6-0) After hydrothermal post-treatment of the titanate nanotubes at 200 $\mathrm{^{\circ}C}$ for more than 1 h, the area of the hysteresis loops at high relative pressures $(P/P_0 > 0.9)$ has an obvious decrease, indicating the disappearance of some large pores ($>50 \text{ nm}$) in the TiO₂ nanofiber samples.

Fig. 6b shows the corresponding pore size distributions of the prepared samples. The pore-size distributions of the $TiO₂$

nanofibers depended on the hydrothermal post-treatment time. Prior to hydrothermal post-treatment, the nanotubes exhibited a wide pore size distribution ranging from 1.5 to more than 100 nm. This was in good agreement with the previous studies though different $TiO₂$ sources were used as the precursors [\[39,40\]. A](#page-6-0)fter hydrothermal post-treatment of the titanate nanotubes at 200 \degree C for 1 h, the pore size distribution of the sample become narrower (1.5–30 nm) and the larger pores (>30 nm) almost disappeared. With further increase in the hydrothermal post-treatment time, the pore size distribution range of the TiO2 nanofibers become further smaller, while the average pore diameter increased from 12.4 to 18.5 nm due to the growth of $TiO₂$ crystallites [\(Fig. 3\).](#page-2-0) Meanwhile, the pore volume of mesopores located at 1.5–10 nm decreased significantly. This may be attributed to the collapse of the tube structure in the nanotubes during the hydrothermal post-treatment. Textural parameters derived from the nitrogen adsorption–desorption isotherm data are summarized in Table 1. When P25 was transformed into the titanate nanotubes, there was a significant increase for S_{BET} from 60.0 to 355.7 m²/g and for pore volume from 0.06 to $1.54 \text{ cm}^3/\text{g}$ due to the formation of nanotube structures. As for the $TiO₂$ nanofibers, with increasing post-treatment time from 1 to 7 h, the *S*_{BET} and pore volume of the samples decreased obviously to $108.2 \text{ m}^2/\text{g}$ and $0.44 \text{ cm}^3/\text{g}$, respectively. According to the above analysis of the pore size distribution curves in Fig. 6b,

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Textural parameters derived from the nitrogen adsorption–desorption isotherm data

^a The BET specific surface area was determined by multipoint BET method suing the adsorption data in the relative pressure (P/P_0) range of 0.05–0.3. b Pore volume and average pore size were determined by nitrogen adsorption

volume at the relative pressure of 0.994.

the decrease of the *S*_{BET} was mainly attributed to the collapse of the tube structure and the growth of $TiO₂$ crystallites, while the obvious decline in the pore volume was attributed to the disappearance of larger pore (>30 nm) and the destruction of nanotube structures. With further increase in the hydrothermal post-treatment time from 7 to 24 h, the specific surface area and pore volume of the samples showed a further decrease. The decreasing value was $18.0 \,\mathrm{m}^2/\mathrm{g}$ for S_{BET} and $0.02 \,\mathrm{cm}^3/\mathrm{g}$ for pore volume, which mainly resulted from the further growth of $TiO₂$ crystallites [\(Fig. 3\).](#page-2-0) However, the S_{BET} (>90.0 m²/g) and pore volume ($>0.40 \text{ cm}^3/\text{g}$) for the obtained TiO₂ nanofibers by this method were obviously larger than those of P25. The high specific surface area and pore volume of the samples are expected to have a wider application in photocatalysis, selective adsorption, ultraviolet blockers, separation, sensing, and as functional filling materials in textile, paints, paper, and cosmetics.

3.4. Photocatalytic activity

The photocatalytic activity of the mesoporous $TiO₂$ nanofibers was evaluated by photocatalytic oxidation of acetone in air. Fig. 7 shows the concentration change of acetone and carbon dioxide with UV illumination time. It could be seen that the concentration of the produced carbon dioxide was about three times higher than that of the destroyed acetone. Further observations indicated that the concentration of acetone and carbon dioxide changed linearly with increasing illumination time, indicating the complete decomposition of acetone [\[41\].](#page-6-0)

The degradation rate constant (or apparent rate constant) of acetone was used to characterize the photocatalytic activity of the mesoporous $TiO₂$ nanofibers. Fig. 8 shows the dependence of apparent rate constants (k, min^{-1}) of the TiO₂ nanofibers on the hydrothermal post-treatment time at 200 ◦C. For comparison, the photocatalytic activity of commercial photocatalyst P25 was also tested under identical conditions. Prior to hydrothermal post-treatment, the titanate nanotubes exhibited no photocatalytic activity for the photocatalytic degradation of acetone in air. After hydrothermal post-treatment of the titanate nanotubes at $200\degree$ C for 1 h, the sample exhibited a decent photocatalytic activity due to the formation of anatase phase. With increasing

Fig. 7. Concentration change of acetone and carbon dioxide with UV illumination time.

Fig. 8. The dependence of apparent rate constants of the $TiO₂$ nanofibers on the hydrothermal post-treatment time.

hydrothermal post-treatment time from 1 to 7 h, the *k* increased significantly and reached a maximum value of 5.18×10^{-3} at 7 h. This was due to the fact that the $TiO₂$ nanofibers showed better crystallization in anatase phase. With further increase in the hydrothermal post-treatment time, the photocatalytic activity of the samples decreased owing to the decrease of the S_{BET} and pore volume ([Table 1\).](#page-4-0) However, as for the $TiO₂$ nanofibers obtained by hydrothermal post-treatment of the titanate nanotubes at 200 \degree C for 3–24 h, the samples showed a higher photocatalytic activity than that of P25. The *k* was determined to be 3.47×10^{-3} for Degussa P25, which is well known to have good photocatalytic activity. The superior activity of $TiO₂$ nanofibers may be ascribed to the fact that the former has larger specific surface area, smaller crystallite size, and higher pore volume. Large specific surface area allows more gaseous reactants to be absorbed onto the surface of the photocatalyst, while high pore volume (mesopores) allows rapid diffusion of various gaseous products during the photocatalytic reaction, which can enhance the rate of photocatalytic reaction. Furthermore, compared with nanosized TiO₂ powder photocatalyst, such as P25, TiO₂ nanofiber photocatalyst can be readily separated from the water in a slurry system after photocatalytic reaction because their lengths are at a scale of several micrometers. Therefore, the obtained $TiO₂$ nanofiber photocatalyst may be suitable for the application at an industrial scale [\[42\].](#page-6-0)

4. Conclusions

Mesoporous anatase $TiO₂$ nanofibers with highly photocatalytic activity were prepared by a simple hydrothermal posttreatment of titanate nanotubes. Hydrothermal post-treatment at $200\degree$ C resulted in the destruction of nanotube structures and the formation of anatase $TiO₂$ nanofibers. The hydrothermal posttreatment time has an obvious effect on the crystallization, crystallite size, morphology, specific surface area, pore structures and photocatalytic activity of the mesoporous $TiO₂$ nanofibers. With increasing hydrothermal post-treatment time, the special surface area and pore volume decreased due to the growth of TiO₂ crystallites. However, the resulted TiO₂ nanofibers could partially maintain the advantage of high specific surface area and pore volume of titanate nanotubes. The photocatalytic activity of the $TiO₂$ nanofibers prepared by this method exceeded that of Degussa P25 when the titanate nanotubes were hydrothermally post-treated at 200 ◦C for 3–24 h. This can be attributed to the fact that the former had larger specific surface area, smaller crystallite size, and higher pore volume.

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